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Nanolithography made from water-based spin-coatable LSMO resist

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Abstract

A dual functional and water soluble spin-coatable lanthanum strontium manganese oxide (LSMO) resist has been developed that consists of lanthanum nitrate, strontium nitrate, manganese nitrate, polyvinyl alcohol and water. Energetic nitrates plus polyvinyl alcohol fuel promote autoignition and produce nanopatterns (<60 nm) upon mild electron beam exposure $(<2 \text{ mC cm}^{-2})$. The formation of cubic perovskite LSMO has been confirmed by micro-IR spectroscopy, elemental analysis, x-ray diffraction and transmission electron microscopy. The patterned LSMO film can be developed using nontoxic and environmentally friendly pure water, and the resist can fabricate active magnetic patterns directly by electron beam exposure. The spin-coatable LSMO resist can be fabricated into either positive or negative patterns easily by varying the electron doses. It can change from negative resist to positive resist and then finally negative resist with the increase of electron dose. The positive and negative dual functional mechanism of spin-coatable LSMO resist is reported. A resist with simultaneous positive and negative capabilities patterning will benefit the direct writing technology of an electron beam. The active magnetic characteristics and high refractive index of the material are useful for the direct fabrication of magnetic and optical devices.

(Some figures in this article are in colour only in the electronic version)

1. Introduction

The ability to manipulate materials at the nanometre scale is a prerequisite of not only studying novel properties of materials at different scales but also of realizing useful miniaturized devices. Conventionally, the fabrication of

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thin film nanostructures is primarily done by using selective etching or templating growth on a prepatterned resist and then performing lift-off. While these approaches lack the ability to control the position of the nanostructure, using an electron beam to create a nanopattern on the material is an easy way to achieve the goal such as has been used for ZnO [1], TiO_2 [2], ZrO_2 [3] and PZT [4]. All of these can make nanoscale negative patterns. Furthermore, a zwitterresist refers

to a resist which, depending on the applied doses, can exhibit both positive and negative properties [5, 6]. A resist with simultaneous positive and negative capabilities of patterning will benefit the direct writing technology of an electron beam.

The solvents used in developing resist are typically volatile and toxic, contributing to health hazards and environmental pollution. The solvents also add to the cost of lithographic processing. Recently, many environmentally friendly lithographic processes have been designed using either a water-based solution [7–12] or supercritical carbon dioxide [13–17] to develop the resist. Here we report a novel pure water developable spin-coatable lanthanum strontium manganese oxide (LSMO) resist. The use of pure water instead of organic or alkaline solvents would undoubtedly be not only environmentally desirable but also could greatly simplify the imaging process.

By partially substituting La³⁺ in functional ceramics LaMnO₃ with divalent ions like Ca²⁺ or Sr²⁺, a number of spectacular properties are generated. The most significant property is the colossal magnetoresistance (CMR) of these manganese oxide materials, and this has stirred up tremendous research interests [18–26]. With the CMR effect, many potential applications, such as field sensors or recording devices, of the manganite thin films are being proposed and explored extensively. In addition, the strontium-doped lanthanum manganite is also known to be a very promising cathode material for a solid oxide fuel cell (SOFC) [24-29] based on stabilized zirconia electrolyte. High-temperature SOFCs have been extensively investigated for the last two decades because of their potential use as clean and efficient power generating devices.

2. Experimental details

The spin-coatable LSMO resist was prepared by dissolving 4.50 wt% of lanthanum nitrate (La(NO₃)₃·6H₂O, Acros, 98%), strontium nitrate, (Sr(NO₃)₂, Riedel-dehaen, p.a.), manganese nitrates (Mn(NO₃)₂·4H₂O, Fluka, >97%) and 1.70 wt% polyvinyl alcohol (PVA, Acros, 88%, 22 000 g mol⁻¹) in water with a molar ratio of La:Sr:Mn = 0.7:0.3:1. The solution was held in a water bath at 25 °C, and stirred for 48 h. The LSMO resist was spin coated at 3000 rpm for 90 s to give a nominal thickness of about 180 nm. High-resolution nanolithography was performed by writing the design pattern across the 150 μ m field with 2.5 nm beam step size in a Hitachi ELS-7500EX machine. A sample containing a 3 × 3 array of field was exposed with a start dose time of 1 μ s and then with the additional dose time increment of 0.01, 0.1 and 1 μ s per field. The exposed sample was developed in pure water for 30 s.

3. Results and discussion

We have developed a direct writing resist from the LSMO using its precursor solution. This solution is functioned as a resist via an autoignition mechanism during electron beam exposure, and the patterned LSMO film can be developed using nontoxic and environmental friendly pure water. Figure 1 shows scanning electron microscopic (SEM) photographs of the finished products made by the positive and negative dual

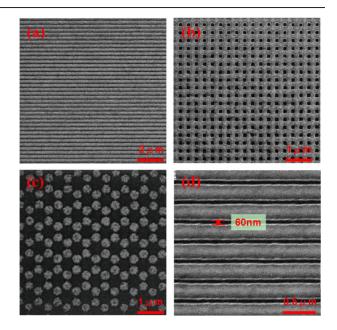


Figure 1. (a) SEM image of patterned lines, 250 nm wide, of spin-coatable LSMO resist. (b) SEM image of a large array of 125 nm lines. (c) SEM image of a periodic pattern with honeycomb style. The diameter of the light-coloured circle in the centre is 400 nm. (d) SEM image of a patterned line: the designed pattern is of straight lines with a width of 240 nm and an interval of 60 nm.

function spin-coatable resist lithography method. Figure 1(a) shows an SEM image of patterned lines, 250 nm wide, of spincoatable LSMO resist. Figure 1(b) shows an SEM image of a large array of 125 nm lines. Figure 1(c) shows an SEM image of a large area periodic pattern with honeycomb style. The diameter of the light-coloured circle in the centre is 400 nm. Figure 1(d) shows an SEM image of a patterned film consisting of spin-coatable LSMO resist: the designed pattern is of straight lines with a width of 240 nm and a interval (distance between lines) of 60 nm. The line width roughness (σ) is about 3.7 nm: this was measured by atomic force microscopy (AFM, Digital Instruments, Dimension-3100 Multimode). Various patterns can be easily fabricated using spin-coatable LSMO resist by electron-beam lithography.

In order to understand the nature of the electron-beam thermal mechanism in the spin-coatable LSMO resist, we removed the water from the LSMO solution to form a gel first then subjected the gel to differential scanning calorimetry (DSC, Du Pont, TA 2910) and thermogravimetry (TGA, Du Pont, TA 2950) analyses. The results are shown in figure 2. The DSC plot shows exothermic peaks ranging from 100 and 150 °C that are due to the crosslinking of the PVA. The TGA curve illustrates a major decomposition at around 223 °C. That indicates the autoignition behaviour as shown by a sharp and nearly vertical step. The result is consistent with an intense exothermic peak at 221 °C in the DSC. This type of decomposition, with a very sharp and intense exothermic reaction, indicates a self-propagation combustion process which has been observed earlier in a few other systems [27, 30-32]. The autoignition of the gel occurs by a thermally induced oxidation-reduction reaction where PVA acts as a fuel and NO_3^- as an oxidant.

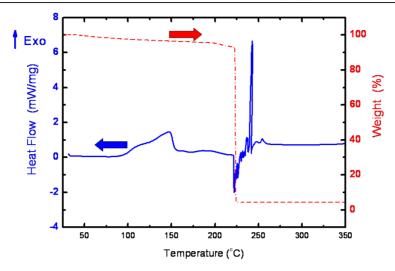


Figure 2. TGA and DSC plot of a spin-coatable LSMO resist.

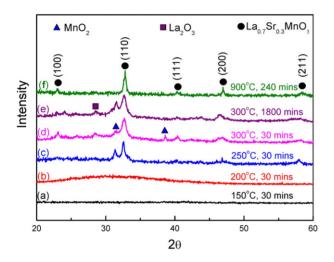


Figure 3. X-ray diffraction patterns of spin-coatable LSMO resists which were calcined at different temperatures.

The autoignition can be induced by mixing an appropriate ratio of the fuel and the oxidant, giving rise to a singlestep decomposition associated with a particular reaction temperature and appreciable exothermic energy so as to facilitate the preparation of LSMO.

To further confirm the formation of LSMO crystalline structure from spin-coatable LSMO resist, we used the x-ray diffraction (XRD) pattern (using a Philips PW1830 diffractometer) of the spin-coatable LSMO resist calcined at high temperature (900 °C) as a control standard. Then we compared it with the XRD patterns obtained from the samples prepared at different temperatures and times. The results are shown in figure 3. The XRD patterns of the samples calcined at 150 and 200 °C, respectively, exhibit an amorphous structure because those temperature are lower than the autoignition temperature of the resist according to our TGA data, whereas the sample calcined at 250 °C shows the formation of the rhombohedral phase of La_{0.7}Sr_{0.3}MnO₃ with traces of MnO₂. The XRD pattern of the sample

 Table 1. Energy dispersive x-ray analysis of elements C and N in the spin-coatable LSMO resist under different conditions.

	Element analysis	
La _{0.7} Sr _{0.3} MnO ₃	Carbon element (wt%)	Nitrogen element (wt%)
E-beam (60 mC cm $^{-2}$)	0.9	0.1
300 °C, 1800 min	2.1	0.3
300 °C, 30 min	3.1	1.6
250 °C, 30 min	4.9	2.0
200 °C, 30 min	12.9	5.4
150 °C, 30 min	19.7	7.3

calcined at 300 °C clearly indicates the formation of the highsymmetry pseudocubic phase of La(Sr)MnO₃ containing trace MnO₂ and La₂O₃ impurity phases. The sample calcined at 900 °C indicates the change from the rhombohedral structure to the pseudocubic form of the perovskite-like structure. The relatively low formation temperature of this phase is attributed to the high reactivity of the homogeneously dispersed ultrafine gel together with the sudden release of thermal energy during the combustion process. This formation temperature is much lower than what has been reported by other investigators employing other chemical methods [27, 33].

An energy dispersive x-ray analysis (EDAX, Philips, XL-30) was used to monitor the amount of the elements C and N in the spin-coatable LSMO resist prepared either with different thermal treatment or with electron-beam lithography treatment. The results are shown in table 1. The elements C and N are from the PVA and metallic nitrates respectively, so they were the indication of the formation of the perovskite phase. The electron-beam treated samples have the least amount of both elements. This result also reveals that the energy from the electron-beam treatment is greater than that from thermal treatment above 300 °C. Therefore, we would obtain a perovskite structure of spin-coatable LSMO resist upon exposure to the electron beam.

The formation of the perovskite phase of the spin-coatable LSMO resist was further investigated by time-resolved Fourier transfer infrared spectra (FTIR, Thermo Nicolet, Nexus 470)

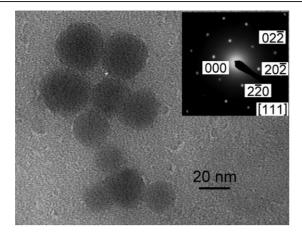


Figure 4. Transmission electron microscopy image of spin-coatable LSMO resist after electron beam exposure. The inset is the selected area diffraction pattern of the patterned LSMO magnetic resist.

acquired in reflectance mode. The absorption peak observed at about 3400 cm⁻¹ is attributed to the hydroxy group of metallic nitrates containing crystalline water. When the calcined temperature was increased, the peak associated with hydroxyl group was reduced, which indicated the formation of LSMO

perovskite phase. The FTIR spectrum of spin-coatable LSMO resist exposed to the electron beam is equivalent to that of the sample calcined at 300 °C.

The transmission electron microscopy (TEM: JEOL, 100CX II) image of the spin-coatable LSMO resist with the electron dose 7.3 mC cm⁻² showed LSMO with a particle size around 30 nm in diameter (figure 4). The selected area electron diffraction image showed a sharp diffraction pattern of the cubic perovskite LSMO with a lattice parameter of 3.8 Å, which matches the XRD results. The spots corresponding to $(2\overline{20})$, $(20\overline{2})$ and $(02\overline{2})$ planes are marked.

The spin-coatable LSMO resist can be fabricated into either positive or negative patterns easily by varying the electron doses, and the feature exhibits dual functions. It can change from negative resist to positive resist and then finally negative resist with the increase of electron dose. Figure 5 illustrates the AFM images of LSMO patterns on Si substrate with different electron dose times. The AFM image of asspin coated sample is illustrated in figure 5(a). The film was very smooth with an RMS roughness of 0.37 nm. Figure 5(b) shows the AFM image of a regular honeycomb array of 100 nm diameter and 500 nm lattice constant LSMO rods at an electron dose of 0.1 mC cm⁻². This step indicates that the spin-coatable LSMO resist can be fabricated into negative patterns due to the crosslinking of the PVA. With

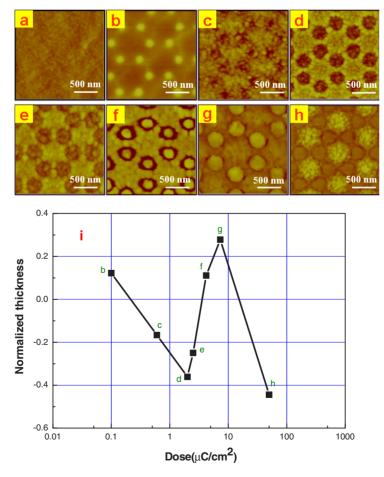


Figure 5. The surface morphology of spin-coatable LSMO resist with different electron dose times. (a) 0 mC cm⁻², (b) 0.1 mC cm⁻², (c) 0.6 mC cm⁻², (d) 2.0 mC cm⁻², (e) 2.5 mC cm⁻², (f) 4.2 mC cm⁻², (g) 7.3 mC cm⁻², (h) 50.0 mC cm⁻², (i) sensitivity curve of spin-coatable LSMO resist. The resist thickness is 180 nm.



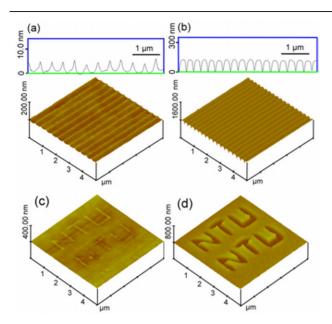


Figure 6. Topographic atomic force microscopy images of dual function spin-coatable LSMO resist. (a) Cross-sectional profiles and AFM images of negative LSMO resist with an electron dose of 0.1 mC cm^{-2} . The height of the grating is 6 nm. (b) Cross-sectional profiles and AFM images of positive LSMO resist with an electron dose of 2 mC cm⁻². The height of the grating is 115 nm. Both negative and positive resist can achieve about 60 nm line width resolution. (c) AFM images of negative LSMO resist with an electron dose of 0.1 mC cm⁻². (d) AFM images of positive LSMO resist with an electron dose of 2 mC cm⁻². Directly writing complex features using LSMO resist can be achieved.

the increase of electron dose, the thermal energy of electron beam will transfer to the neighbouring area and make all of the PVA crosslink. As illustrated in figure 5(c), the PVA starts to decompose at the direct beam location. Finally, the spin-coatable LSMO resist becomes a positive pattern with an electron dose of 2 mC cm⁻² (figure 5(d)). The RMS roughness of the spin-coatable LSMO illustrated in figure 5(d) is about 7.8 nm. Above this threshold electron-beam thermally energy, the increase of electron dose starts the autoignition process of LSMO. Figures 5(e), (f) and (g) show the AFM images of a regular honeycomb array with the electron dose of 2.5 4.2 and 7.3 mC cm⁻², respectively. With an electron dose higher than 50 mC cm⁻², the LSMO honeycomb patterns eventually collapsed, as illustrated in figure 5(h). Figure 5(i) illustrates the sensitivity curves of the spin-coatable LSMO resist after electron beam exposure and development.

Both the negative and positive properties of spin-coatable LSMO resist can achieve nanoscale resolution. The 3D AFM topographic images and cross-sectional profiles of a section of negative and positive spin-coatable LSMO resist with 60 nm line width corresponding to an electron dose of 0.1 mC cm^{-2} and 2 mC cm⁻² respectively are shown in figures 6(a) and (b). The period of the fabricated grating was 300 nm. It is obvious from the photograph that the heights of gratings are uniform. This dual function spin-coatable LSMO resist is also capable of directly patterning large and complex features by controlling the dose time. Figure 6(c) shows the AFM image of negative spin-coatable LSMO resist features that were written at an

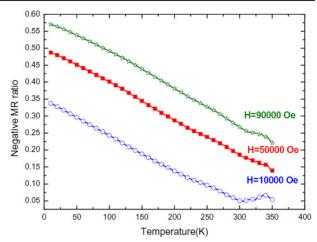


Figure 7. The relationship of the negative magnetoresistance (MR) ratio and the temperature for the spin-coatable LSMO resist calcined at 900 °C under different external magnetic field strengths.

electron dose of 0.1 mC cm⁻², and figure 6(d) shows the positive spin-coatable LSMO resist features that were written at an electron dose of 2 mC cm⁻². The results illustrate that the capability of spin-coatable LSMO resist can be either positive or negative by varying the electron doses. The development of spin-coatable LSMO resist is an important research topic in the study of a new magnetic resist with both positive and negative resist properties, water-soluble solvent developable property, and the capability of preparing nanoscale patterns.

All the samples for the refractive index and film thickness measurements were deposited on a silicon wafer. The refractive index and film thickness were determined by spectral microreflectometry (Mission Peaks Optics, MP100-M), by measuring the interference between incident and reflected light with wavelengths ranging from UV to visible (250–1000 nm). From the phase shift, the film thickness and refractive index (n and k) can be determined. The film thickness of spin-coatable LSMO resist approaches 180 nm, and the refractive index is about 2.38.

In order to study the magnetic properties of spin-coatable LSMO resist, the resist sample was calcined at 900°C for 4 h first, then the magnetoresistance of the sample was evaluated by a physical properties measurement system (PPMS, PPMS-9, Quantum Design). The relationship between the temperature and the resistance of the samples under different external magnetic field strengths was obtained in order to understand the influence of the external magnetic field on the resistance. External magnetic field strengths of 10000, 50000, and 90000 Oe were selected to observe the changes in the resistance under different external magnetic field strengths. The resistance of spin-coatable LSMO resist increases gradually as the external magnetic field strength increases, as shown in figure 7. Therefore, the influences of the external magnetic field on the spin-coatable LSMO resist samples show negative and colossal magnetoresistance effects.

4. Summary

We have presented a novel positive and negative dual function water-based magnetic resist material. The water-

based material is nontoxic and environmental friendly. The resist material consists of the oxide precursors of La, Sr, Mn, and a phase formation promoter. The phase formation promoter is polymer that assists the formation of the perovskite crystalline structure of lanthanum strontium manganese oxide. Pure water is used to develop the resist. Various nanoscale patterns can be formed from the material using an electron beam. An external magnetic field on the spin-coatable LSMO resist samples shows colossal and negative magnetoresistance effects. This electron-beam writable LSMO resist with active magnetic characteristics and high refractive index (~ 2.38) is useful for the direct fabrication of magnetic and optical device.

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References

- Saifullah M S M, Subramanian K R V, Kang D-J, Anderson D, Huck W T S, Jones G A C and Welland M E 2005 Adv. Mater. 17 1757
- [2] Saifullah M S M, Subramanian K R V, Tapley E, Kang D-J, Welland M E and Butler M 2003 *Nano Lett.* 3 1587
- [3] Subramanian K R V, Saifullah M S M, Tapley E, Kang D-J, Welland M E and Butler M 2004 Nanotechnology 15 158
- [4] Alexe M, Harnagea C, Visinoiu A, Pignolet A, Hesse D and Gösele U 2001 Scr. Mater. 14 1175
- [5] Chen J K, Ko F H, Chen H L and Chang F C 2003 Japan. J. Appl. Phys. 42 3838
- [6] Chen J K, Ko F H and Chang F C 2005 Adv. Funct. Mater. 15 1147
- [7] Lin Q, Steinhäusler T, Simpson L, Wilder M, Medeiros D R, Willson C G, Havard J and Fréchet J M J 1997 *Chem. Mater*. 9 1725

- [8] Havard J M, Shim S Y, Fréchet J M J, Lin Q, Medeiros D R, Willson C G and Byers J D 1999 Chem. Mater. 11 719
- [9] Havard J M, Vladimirov N, Fréchet J M J, Yamada S, Willson C G and Byers J D 1999 Macromolecules 32 86
- [10] Havard J M, Yoshida M, Pasini D, Vladimirov N, Fréchet J M J, Medeiros D R, Patterson K, Yamada S, Willson C G and Byers J D 1999 J. Polym. Sci. A 37 1225
- [11] Chae K H, Sun G J, Kang J K and Kim T K 2002 J. Appl. Polym. Sci. 86 1172
- [12] Yamada S, Mrozek T, Rager T, Owens J, Rangel J, Willson C G and Byers J 2004 Macromolecules 37 377
- [13] Sundararajan N, Yang S, Ogino K, Valiyaveettil S, Wang J G, Zhou X Y, Ober C K, Obendorf S K and Allen R D 2000 *Chem. Mater.* 12 41
- [14] Cooper A I 2001 Adv. Mater. 13 1111
- [15] Pham V Q, Ferris R J, Hamad A and Ober C K 2003 Chem. Mater. 15 4893
- [16] Weibel G L and Ober C K 2003 Microelectron. Eng. 65 145
- [17] Mao Y, Felix N M, Nguyen P T, Ober C K and Gleason K K 2004 J. Vac. Sci. Technol. B 22 2473
 [10] Z. Global Phys. Rev. Lett. 101
- [18] Zener C 1951 *Phys. Rev.* **81** 440
- [19] Jin S, Tiefel T H, McCormack M, Fastnacht R A, Ramesh R and Chen L H 1994 Science 264 413
- [20] Mahesh R, Mahendiran R, Raychaudhuri A K and Rao C N R 1996 Appl. Phys. Lett. 68 2291
- [21] Wu Y, Suzuki Y, Rudiger U, Yu J, Kent A D, Nath T K and Eom C B 1999 Appl. Phys. Lett. 75 2295
- [22] Fontcuberta J, Bibes M, Martinez B, Trtik V, Ferrater C, Sanchez F and Varela M 1999 J. Appl. Phys. 85 4800
- [23] Yang S Y, Kuang W L, Ho C H, Tse W S, Lin M T, Lee S F, Liou Y and Yao Y D 2001 J. Magn. Magn. Mater. 226 690
- [24] Haghiri-Gosnet A M and Renard J P 2003 J. Phys. D: Appl. Phys. 36 R127
- [25] Lu H B et al 2004 Appl. Phys. Lett. 84 5007
- [26] Geck J, Wochner P, Kiele S, Klingeler R, Reutler P, Revcolevschi A and Buchner B 2005 *Phys. Rev. Lett.* 95 236401
- [27] Basu R N, Pratihar S K, Saha M and Maiti H S 1997 Mater. Lett. 32 217
- [28] Huang K Q and Goodenough J B 2000 J. Alloys Compounds 303 454
- [29] Lee H K 2002 Mater. Chem. Phys. 77 639
- [30] Chakraborty A, Devi P S, Roy S and Maiti H S 1994 J. Mater. Res. 9 986
- [31] Gaudon M, Laberty-Robert C, Ansart F, Stevens P and Rousset A 2002 Solid State Sci. 4 125
- [32] Antony S A, Nagaraja K S, Reddy G L N and Sreedharan O M 2001 Mater. Lett. 51 414
- [33] Majewski P, Epple L, Rozumek M, Schluckwerder H and Aldinger F 2000 J. Mater. Res. 15 1161