Polarization-dependent confocal Raman microscopy of an individual ZnO nanorod

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In this study, polarized-Raman scattering measurements of an individual ZnO nanorod were carried out by using a confocal microscope together with a high-resolution piezoelectric stage. A predominant A_1 (TO) mode at 378 cm⁻¹ in the parallel polarization (E_{\parallel}) configuration and a predominant E_2 (high) mode at 438 cm⁻¹ in the perpendicular polarization (E_{\perp}) configuration demonstrate the strong polarization dependent Raman scattering signals of an individual ZnO nanorod. The Raman intensity images of the individual ZnO nanorods with different orientation configurations can also be obtained, which reflect the interplay between the local phonon behavior and geometric anisotropy. © 2008 American Institute of Physics. [DOI: 10.1063/1.2938701]

ZnO has a direct band gap of 3.37 eV with a large exciton binding energy of 60 meV at room temperature. Due to the strong binding energy of excitons, ZnO-based semiconductors are recognized as very promising materials for micro- or nanooptoelectronic applications in the UV region.¹⁻⁵ One-dimensional nanometer-sized semiconductor materials, i.e., nanowires and nanorods, have recently attracted considerable attention due to consisting of a high surface-to-volume ratio, which can critically affect the electronic and optical properties.^{6–8} Large optical anisotropy in the well-aligned ZnO nanowires have been demonstrated, due to its high aspect ratio nature of the one-dimensional nanomaterial.⁹ Raman scattering is a very useful nondestructive measurement to probe the phonon behavior of materials, providing the information of crystal structure, lattice dynamics, and defects. Recently, investigations of Raman scattering on an individual GaN nanowire,^{10,11} nanorod,¹² carbon nanotube,¹³ SiC nanowire,¹⁴ and CdSe nanowire¹⁵ have been demonstrated. In this article, polarization-dependent Raman scattering measurements on individual ZnO nanorods were investigated by confocal Raman microscopy together with a high-resolution piezoelectric stage. Mapping of the Raman shift signal was also conducted to probe the local information of phonon behavior of an individual ZnO nanorod, correlating with its geometric and orientation-dependent anisotropy.

ZnO nanorods were grown on seeded indium tin oxide (ITO) substrates using chemical bath deposition. Formation of the seed layer on ITO was conducted by the dip-coating method using an aqueous solution of zinc acetate dihydrate $(Zn(CH_3COO)_2 \cdot 2H_2O, ZnAc \cdot 2H_2O)$ and hexamethylenetetramine $(C_6H_{12}N_4, HMTA)$. The scanning electron microscope (SEM) images of the as-grown ZnO nanorod array and an individual ZnO nanorod are shown in Figs. 1(a) and 1(b), respectively. The ZnO nanorods consist of a length of about 10 μ m and diameters ranging from 180 to 300 nm, which were obtained by multiple bath growth in a fresh aqueous

solution of ZnAc·2H₂O and HMTA at 95 °C for every 3 h. The high-resolution transmission electron microscope (TEM) image of an individual ZnO nanorod and the corresponding selection area electron diffraction pattern are also shown in the Fig. 1(c). They reveal that the nanorod possesses the single-crystal structure and the lattice spacing of around 0.52 nm along the longitudinal axis direction (*c* axis) corresponds to the *d* spacing of ZnO (001) crystal planes.

Raman scattering measurements were conducted with a confocal microscopy (WITec, CMR200, Germany) in the backscattering mode. The scanning confocal Raman images were collected using a high-resolution piezoelectric stage (PI). A He–Ne laser (λ_{exe} =632.8 nm) is used as the excitation source. The polarization of the light was selected using a half-wave plate. The Raman image mapping was obtained by intensity integration of the spectra by recording the data with 200 nm/step and an integration time of 1 s/step. In all cases, the laser beam was focused down with a 100× numerical aperture=0.95 objective (Olympus) and the focused laser beam was about 1 μ m in diameter, corresponding to about 4–5 pixels in the integrated Raman image.

According to the group theory, four Raman active modes of A_1 , E_1 , and $2E_2$ (E_2^{low} and E_2^{high}) are expected for the wurtzite-type ZnO structure, which belongs to the space group of C_{6v}^4 .¹⁶ In addition, the polar nature of A_1 and E_1 modes leads to the splitting of phonon signals into transverse-optical (A_{1T} and E_{1T}) and longitudinal optical (A_{1L} and E_{1L}) components. For the individual ZnO nanorod, three different polarization configurations of $\overline{X}(Y,Y)X$, $\overline{X}(Z,Z)X$, and $\overline{X}(Y,Z)X$ according to the Porto notation were investigated, where the *c* axis of the nanorod is set as the *Z* axis. The term of $\overline{X}(Y,Z)X$ represents that the laser light is incident from the opposite *X* direction with the *Y* polarization, while the scattered signal is collected with the *Z* polarization in the *X* direction. The coordination of the single ZnO nanorod with respect to the incident/scattered light is shown in Fig. 2(a). Figure 2(b) shows the Raman shift spectra of an individual ZnO nanorod with different polarization configurations. For comparison, the Raman signal of the as-grown ZnO nanorod array (denoted as *ensemble*) was also mea-

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FIG. 1. (a) The SEM micrograph of the well-aligned ZnO nanorods. The scale bar is 6 μ m. (b) The SEM image of an individual ZnO nanorod. The scale bar is 100 nm. (c) High-resolution TEM image of an individual ZnO nanorod (growth along the *c* axis) and the corresponding electron diffraction pattern.

sured. For the as-grown sample, the peaks at 331, 378, 410, and 438 cm⁻¹ can be resolved, which can be assigned to A_1 (acoustic overtone), A_1 (TO), E_1 (TO), and E_2 (high) symmetry modes, respectively, which are consistent with those that reported the Raman shift signals of the bulk ZnO thin films or ensemble of nanostructures.^{17–19} For the Raman shift signals of the individual ZnO nanorod, strong polarization dependence on Raman signatures can be found. In the $\overline{X}(Y, Y)X$ configuration, where the incident/scattered photons



FIG. 2. (Color online) (a) The coordination of the single ZnO nanorod with respect to the incident/scattered light. $\theta(\varphi)$ is the angle between the laser polarization and the Z axis (X axis). (b) The Raman shift spectra of the single ZnO nanorod with different polarization configurations of $\overline{X}(Y,Y)X$, $\overline{X}(Z,Z)X$, and $\overline{X}(Y,Z)X$ according to the Porto notation. The spectra of the ensemble represents the Raman scattering spectra of the as-grown ZnO nanorod array.

are polarized perpendicular to the c axis of the nanorod (denoted as E_{\perp}), the peak at 438 cm⁻¹ of the E_2 (high) mode becomes dominant, accompanying a very weak intensity of A_1 (TO) mode at 378 cm⁻¹ and a negligible signal of the peak at 331 cm⁻¹. The relative intensity of the individual peaks in the E_{\perp} configuration is similar to the Raman shift signatures of the ZnO nanorod array, which were grown along the *c* axis. In contrast, for the $\overline{X}(Z,Z)X$ configuration, where the incident/scattered light has the polarization parallel to the c axis of the nanorod (denoted as E_{\parallel}), the peak at 378 cm^{-1} corresponding to the A_1 (TO) mode becomes dominant with a large suppression of the E_2 (high) mode at 438 cm⁻¹. In addition, the peak of A_1 (acoustic overtone) at 331 cm⁻¹ also persists well in the E_{\parallel} polarization condition. The Raman intensity I can be usually expressed as $I \sim |\hat{e}_s \cdot \vec{R} \cdot \hat{e}_l|^2$, where \vec{R} is the Raman tensor and $e_l = \cos \varphi \hat{X}$ $+\sin\varphi\sin\theta\hat{Y}+\sin\varphi\cos\theta\hat{Z}, \quad e_s=\cos\varphi\hat{X}+\sin\varphi\sin\beta\hat{Y}+\sin\varphi$ $\varphi \cos \beta \hat{Z}, \hat{e}_{l}(\hat{e}_{s})$ is the unit vector of the incident (scattered) light polarization and θ (or β) is the angle between the incident (or scattered) light polarization and the nanorod axis. In our case, $\phi = \pi/2$ corresponds to the angle between the propagating direction of scattered light and the orientation of the nanorod axis. According to the theoretical prediction,^{11,12} when $\theta = \beta$, the intensity of the A_1 (TO) at 378 cm⁻¹ should become dominant when $\theta = 0^{\circ}$ and the E_2 (high) mode at 438 cm⁻¹ should be dominant when $\theta = 90^{\circ}$, which shows a good consistency with the above observation in the X(Z,Z)Xand X(Y, Y)X configurations, respectively. However, the nonzero signal at 438 cm⁻¹ (or 378 cm⁻¹) in the $\overline{X}(Y, Y)X$ (or $\overline{X}(Z,Z)X$ configuration is mainly attributed to nonperfect orientation of the nanorod with respect to the polarization of

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FIG. 3. (Color online) (a) Raman scattering mapping image of two individual ZnO nanorods with orientation almost perpendicular to each other. The inset in (a) shows the optical image of ZnO nanorods. (b) Raman images generated by integration of $(370-385 \text{ cm}^{-1})$ and (c) $(430-445 \text{ cm}^{-1})$ spectral ranges for the $X(Z,Z)\overline{X}$ polarization configuration.

the incident/scattered light. As the polarization configuration is changed to $\overline{X}(Z, Y)X$, all the A_1 (TO), E_1 (TO), and E_2 (high) modes appear. The above result clearly accounts for the nature of polarization-dependent anisotropy in the Raman spectra of an individual ZnO nanorod.

We have further chosen two individual ZnO nanorods with orientations almost perpendicular to each other to perform the Raman scattering image mapping experiment. The inset in Fig. 3(a) shows the optical microscopic image of the two nanorods. By intensity integration of the spectral domain ranging from 250 to 900 cm⁻¹, the Raman image of the two individual ZnO nanorods, labeled as NR 1 (nanorod 1) and NR 2 (nanorod 2), can be obtained, as shown in Fig. 3(a). By integrating the intensity of the spectral domain $(370-385 \text{ cm}^{-1})$ around the A_1 (TO) mode at 378 cm⁻¹ and of the spectral domain (430–445 cm⁻¹) around the E_2 (high) mode at around 438 cm⁻¹ in the $\overline{X}(Z,Z)X$ configuration, the complementary images with the maximum signals almost on the portion of the NR1 and NR2 are obtained, as shown in Figs. 3(b) and 3(c), respectively. The polarization of incident light is now pointing along the scale bar, as shown in Fig. 3(a), which is assigned as the Z axis. The result provides the information correlating the local phonon behavior with geometric and orientation-dependent anisotropy. However, the nonvanishing Raman intensity image at the two ends of the NR1 in Fig. 3(c) is possibly due to (i) the nonperfect orientation of the NR1 or nonuniform crystal growth along the caxis with respect to the incident photon polarization or (ii) the boundary mismatch of the dielectric function ε between a nanorod and its surroundings (air), which may also cause the strong dependence of the transmission of the light inside a nanorod on its shape, orientation and size.^{14,15,20}

In summary, an individual ZnO nanorod has demonstrated a large anisotropy in its Raman scattering spectra when different incident/scattered light polarization configurations are used. The Raman image mapping signals provide direct information accounting for the anisotropic phonon behavior of the individual ZnO nanorods.

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